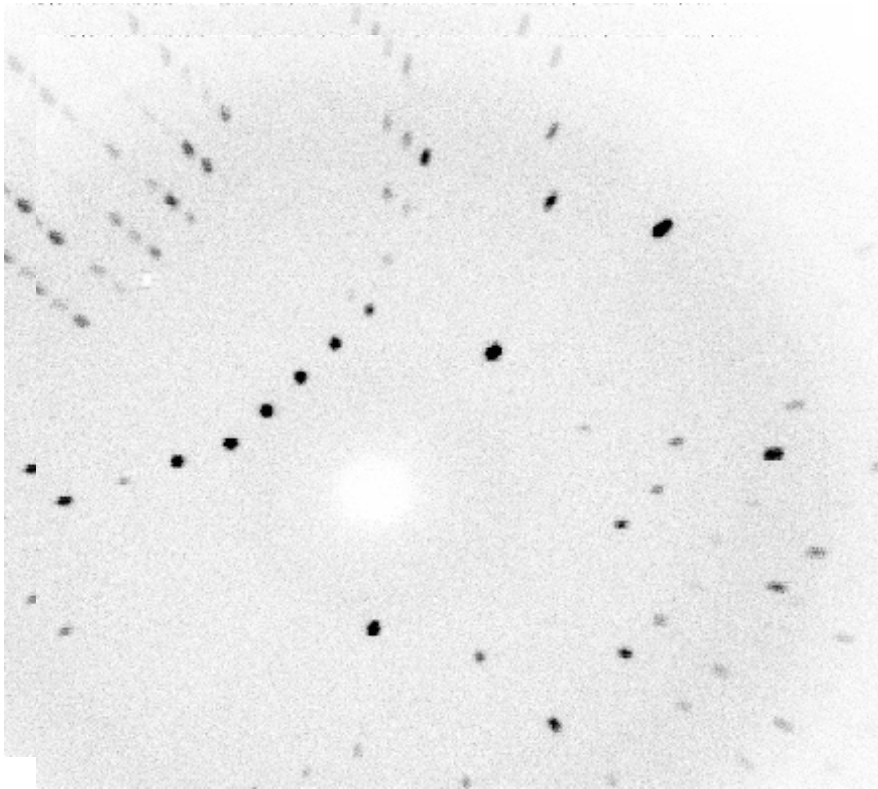
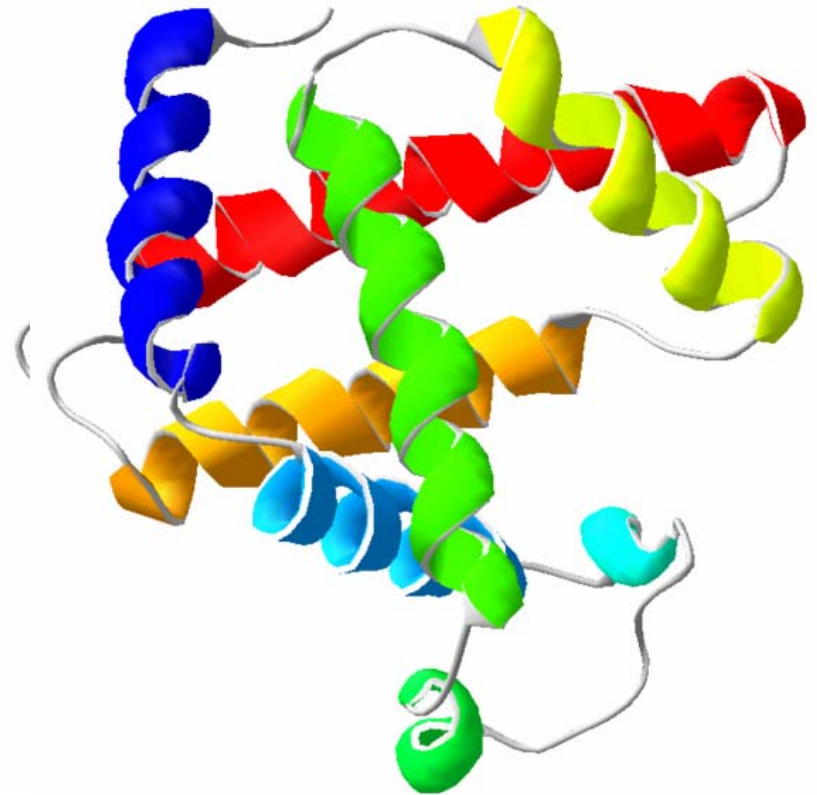


# 3.012 Fund of Mat Sci: Structure – Lecture 18

## X-RAYS AT WORK



An X-ray diffraction image for the protein myoglobin. Source: Wikipedia.



Model of helical domains in myoglobin. Image courtesy of Magnus Manske

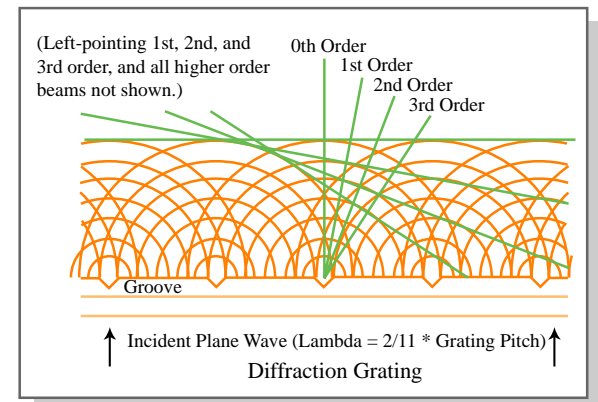
# Homework for Wed Nov 23

- Prof Wuensch Lecture Notes
- <http://capsicum.me.utexas.edu/ChE386K/> for many details (Lect 19 onwards, but note different  $2\pi$  convention)
- Buy turkey

# Last time:

1. X-rays generation: undulators and wigglers in synchrotrons, bremsstrahlung and core excitations (e.g.  $K_{\alpha}$ ) in X-ray tubes
2. Reciprocal lattice
3. Diffraction gratings – Huygens construction
4. Laue diffraction from periodic arrays in 1-d, 2-d, 3-d

Figure by MIT OCW.



# Reciprocal lattice (IV)

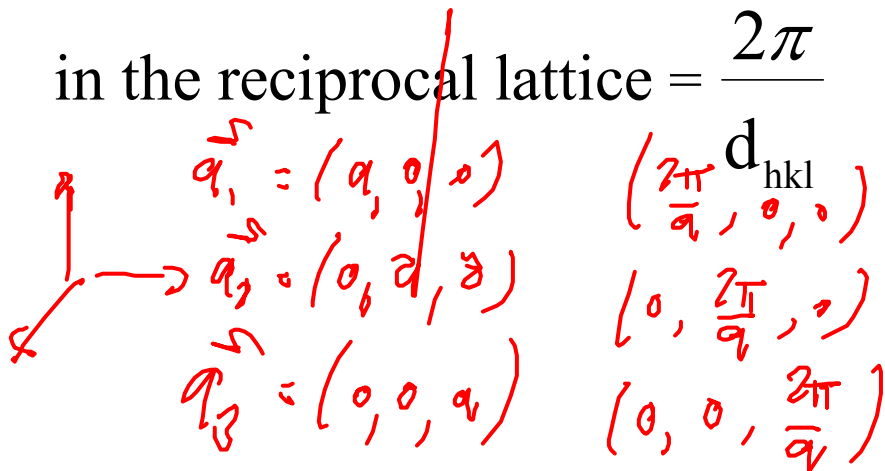
$$\vec{G} = h\vec{b}_1 + i\vec{b}_2 + j\vec{b}_3 \quad \text{with } h, i, j \text{ integers,}$$

$$\vec{b}_1 = 2\pi \frac{\vec{a}_2 \times \vec{a}_3}{\vec{a}_1 \cdot (\vec{a}_2 \times \vec{a}_3)} \quad \vec{b}_2 = 2\pi \frac{\vec{a}_3 \times \vec{a}_1}{\vec{a}_2 \cdot (\vec{a}_3 \times \vec{a}_1)} \quad \vec{b}_3 = 2\pi \frac{\vec{a}_1 \times \vec{a}_2}{\vec{a}_3 \cdot (\vec{a}_1 \times \vec{a}_2)}$$

$\vec{G} = (h, i, j)$  are the reciprocal-lattice vectors

$d_{hkl}^*$  is distance between two planes of Miller indices  $h \ k \ l$

in the reciprocal lattice =  $\frac{2\pi}{d_{hkl}}$



$\vec{a}_1 = (a_1, 0, 0) \quad \left(\frac{2\pi}{a_1}, 0, 0\right)$   
 $\vec{a}_2 = (0, a_2, 0) \quad \left(0, \frac{2\pi}{a_2}, 0\right)$   
 $\vec{a}_3 = (0, 0, a_3) \quad \left(0, 0, \frac{2\pi}{a_3}\right)$

$\vec{a}_i \cdot \vec{b}_j = 2\pi \delta_{ij}$   
 $e^{i\vec{k} \cdot \vec{r}} \mapsto e^{i\vec{k} \cdot (\vec{r} + \vec{R})}$

# First and second Laue conditions

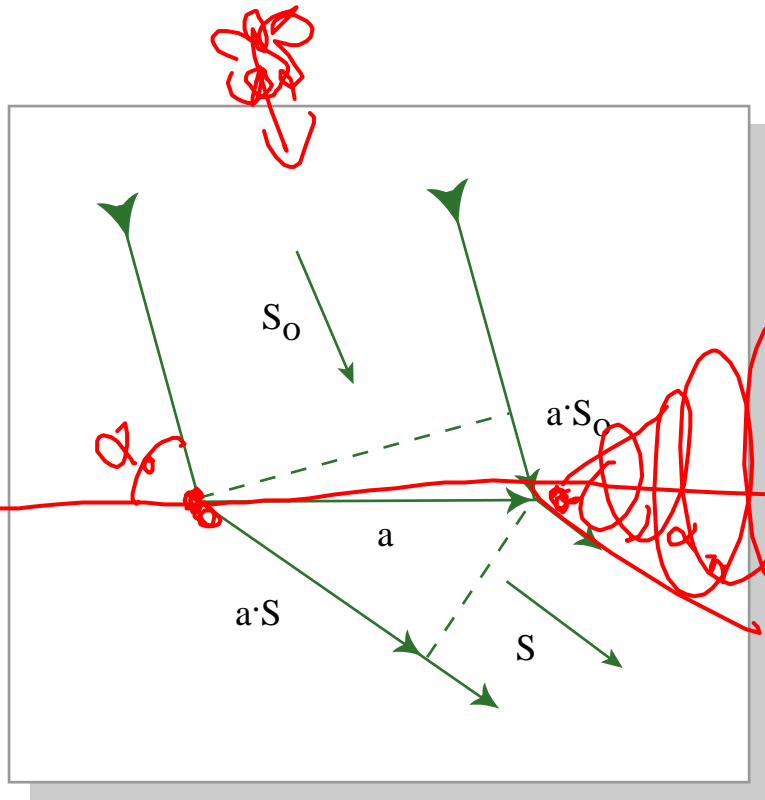


Figure by MIT OCW.

$$\vec{a} \cdot \vec{S} = a \cos \alpha_n$$

$$\vec{a} \cdot \vec{S}_0 = a \cos \alpha_0$$

$$\vec{a} (\cos \alpha_n - \cos \alpha_0) = \vec{a} \cdot (\vec{S} - \vec{S}_0) = n_x \lambda$$

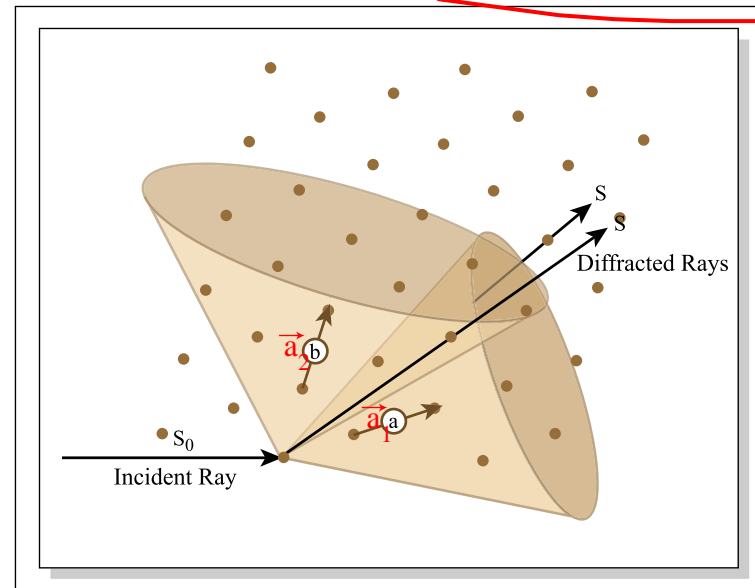


Figure by MIT OCW.

# All three Laue conditions

$$\vec{a}_1 \cdot (\vec{S} - \vec{S}_0) = \text{integer multiple of } \lambda$$

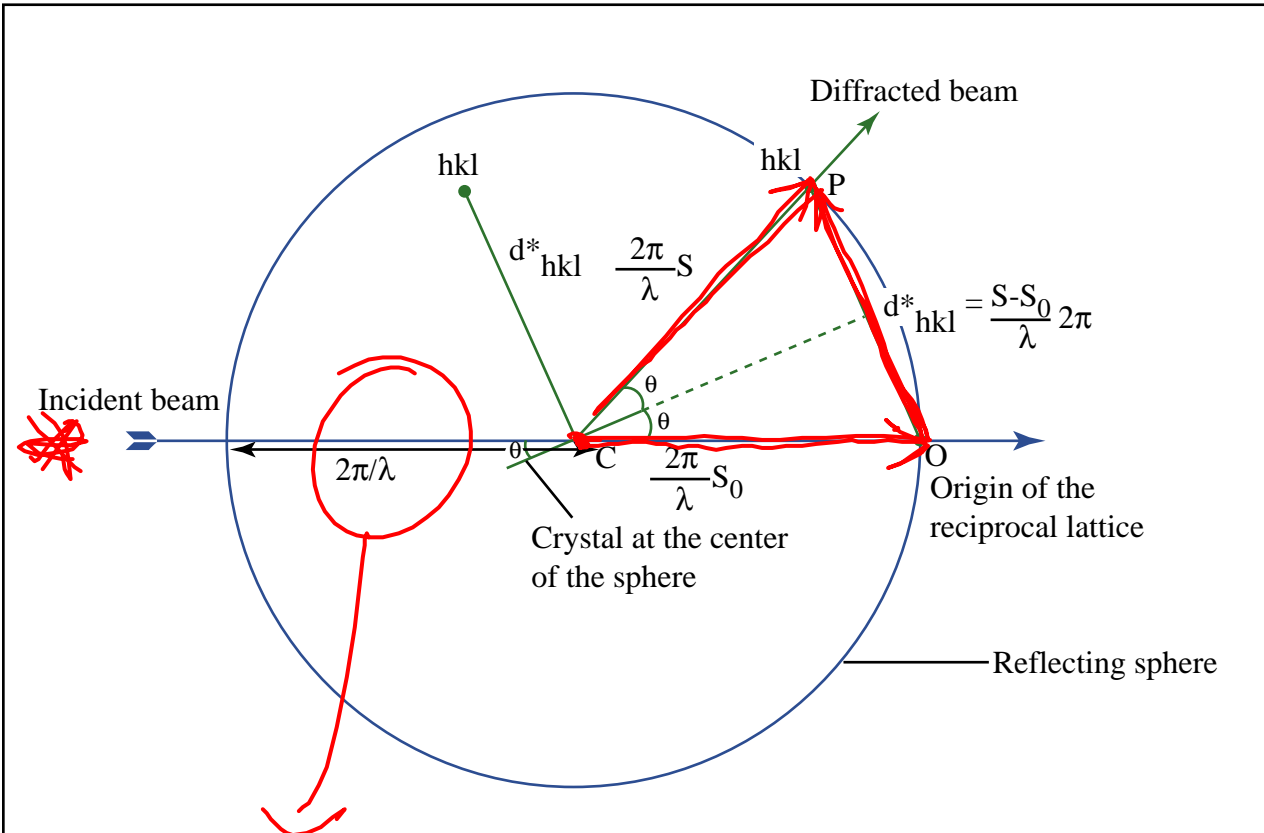
$$\vec{a}_2 \cdot (\vec{S} - \vec{S}_0) = \text{integer multiple of } \lambda$$

$$\vec{a}_3 \cdot (\vec{S} - \vec{S}_0) = \text{integer multiple of } \lambda$$

$$\vec{a}_i \cdot \left( \frac{\Delta \vec{S}}{\lambda} \right) = 2\pi \text{ INTEGER}$$

$\frac{\Delta \vec{S}}{\lambda} \in \text{RECIPROCAL LATTICE}$

# Ewald construction



$$\frac{\Delta S}{\lambda} 2\pi \in \text{RECIP. LATT.}$$

RADIUS  $\frac{2\pi}{\lambda}$

Figure by MIT OCW.

# Laue condition needs “white” spectrum

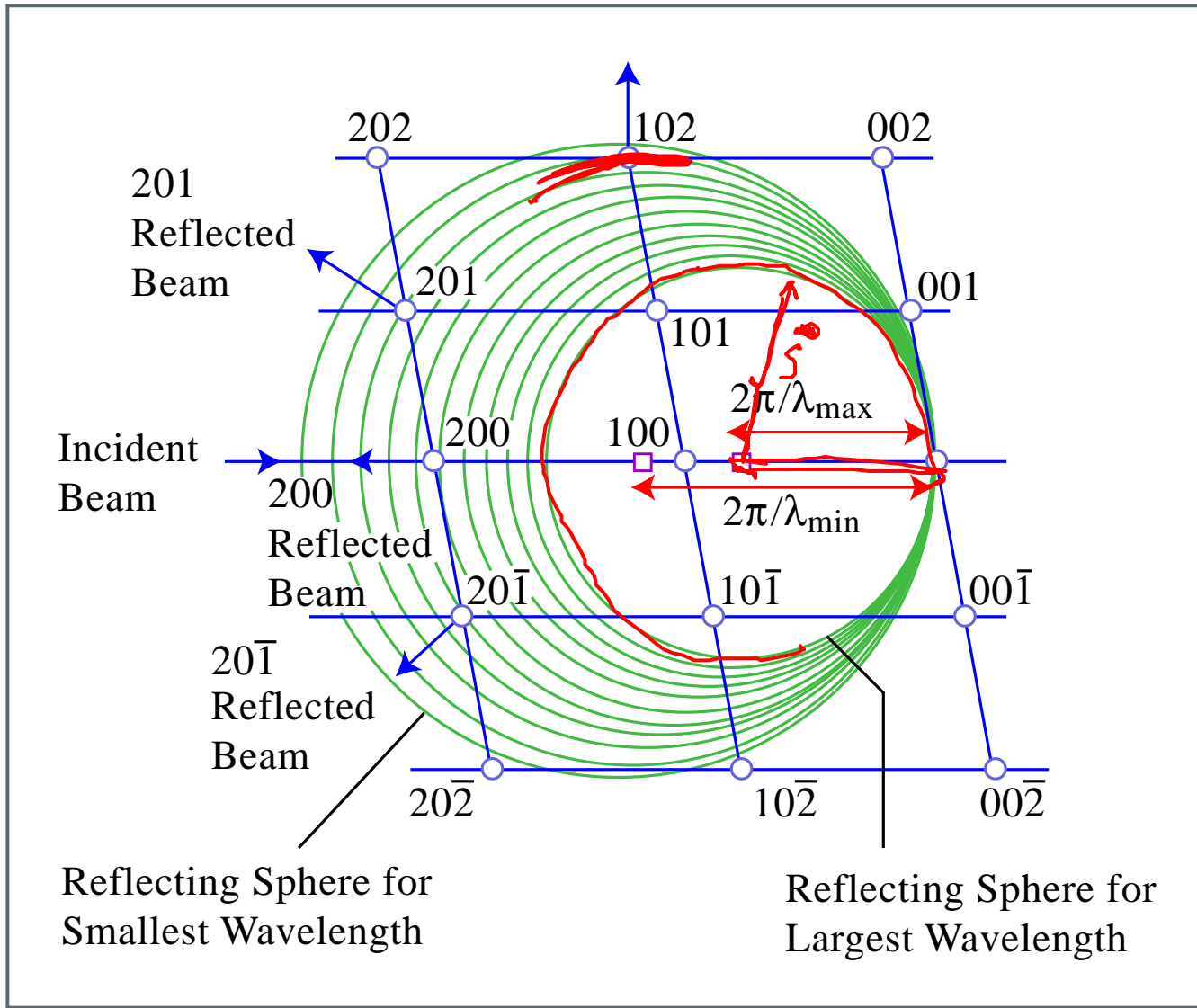


Figure by MIT OCW.



# Alternate geometrical view

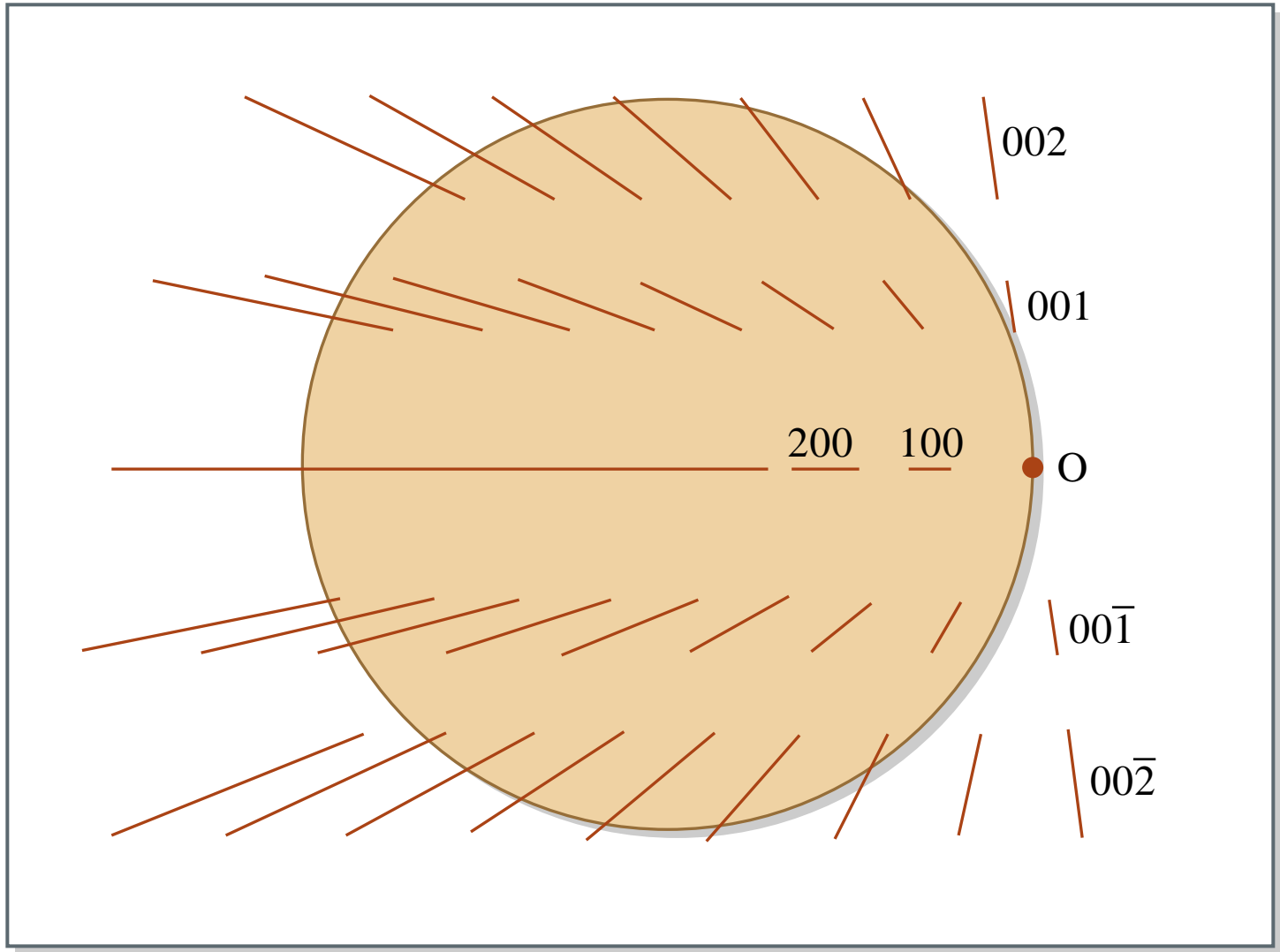


Figure by MIT OCW.

# Bragg Law

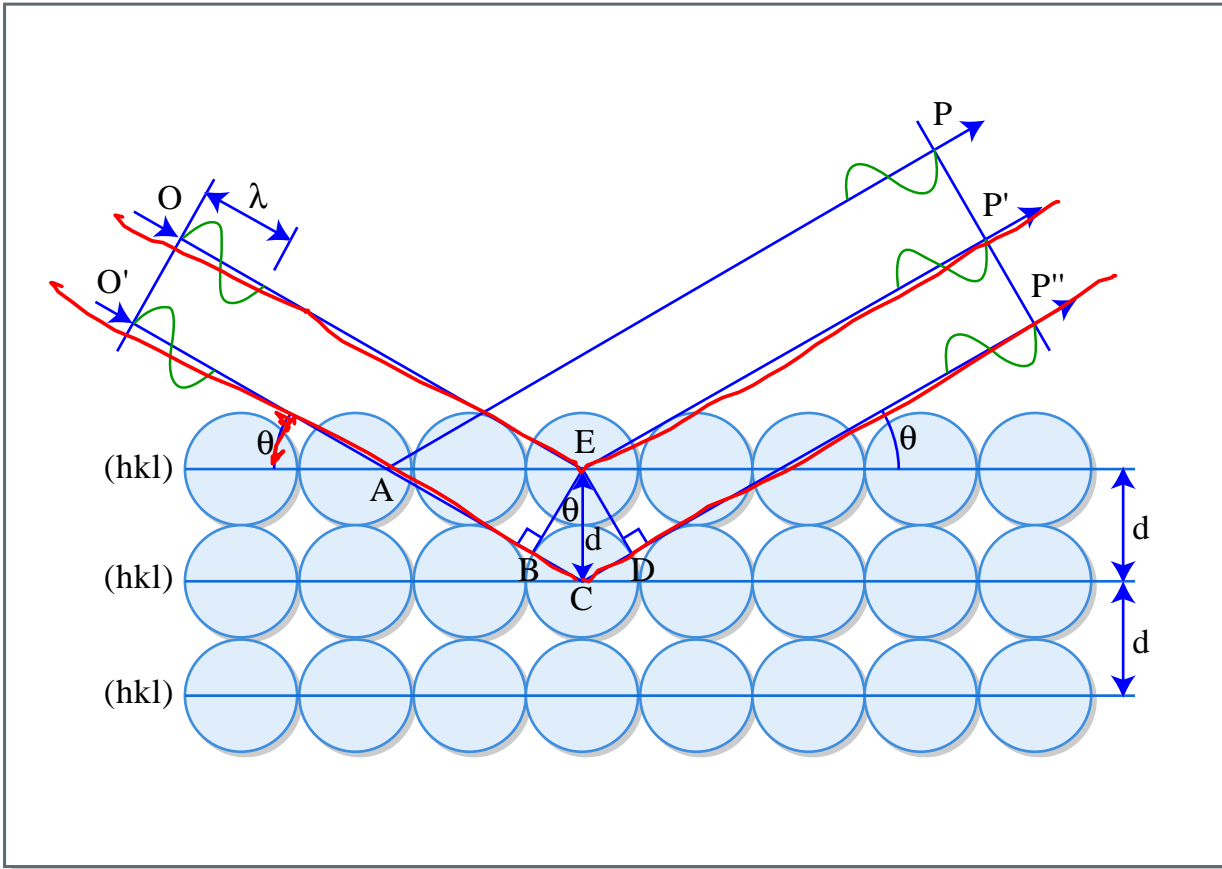


Figure by MIT OCW.

$$n\lambda = d_{hkl} 2 \sin \theta$$

# Equivalence to Laue condition

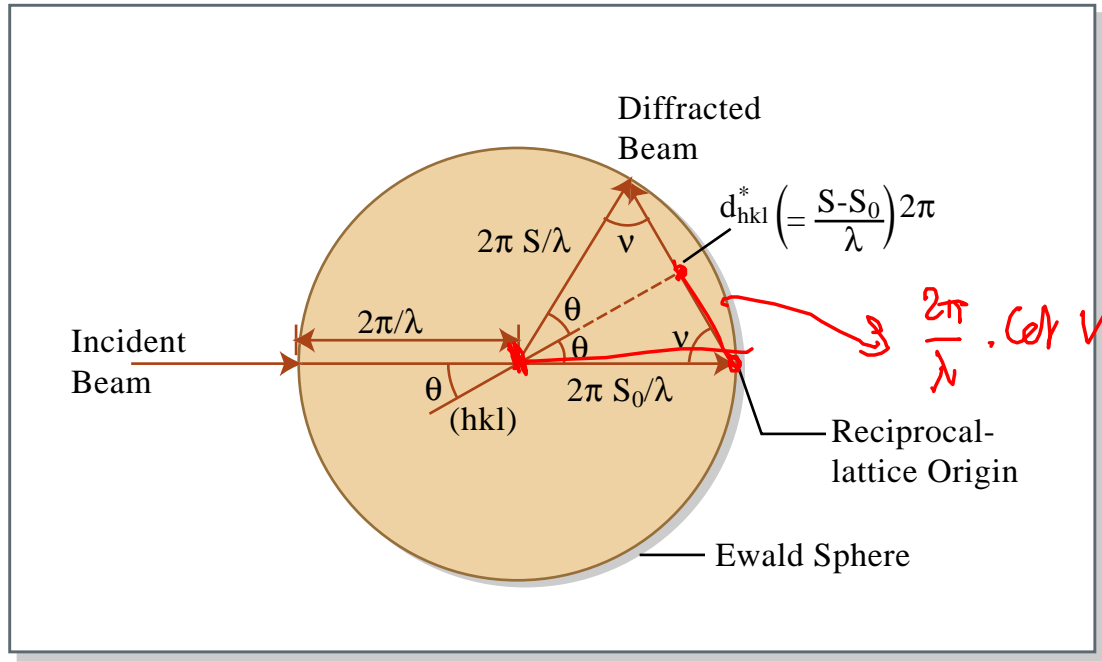


Figure by MIT OCW.

$\lambda = 2 d_{hkl} \sin \theta$

$$2\pi \left( \frac{\vec{S} - \vec{S}_0}{\lambda} \right) = \frac{2\pi}{\lambda} \cos \nu = d_{hkl}^* = \frac{2\pi}{d_{hkl}} = \frac{2\pi}{\lambda} 2 \sin \theta$$

# Powder diffraction (I)

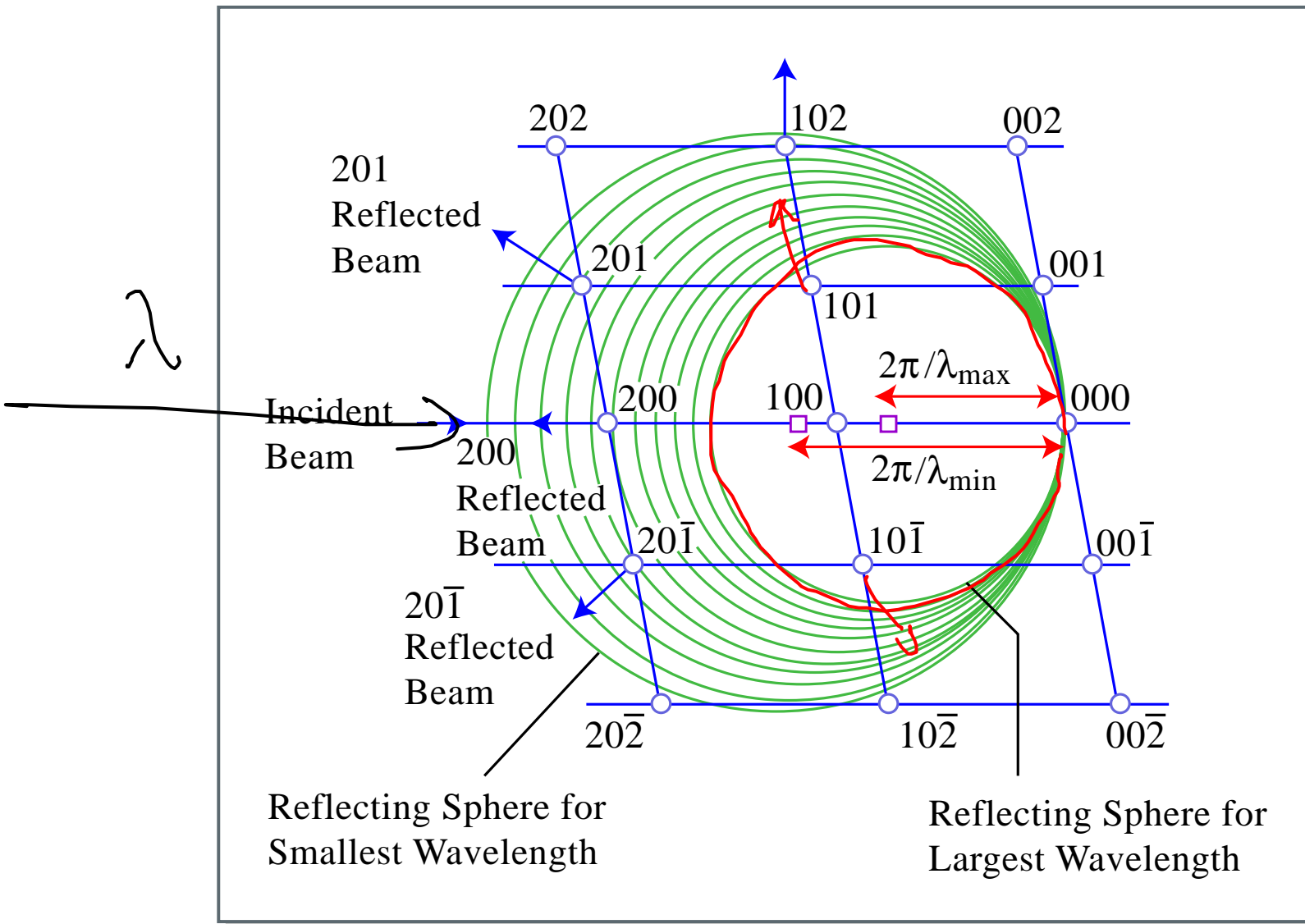


Figure by MIT OCW.

# Powder diffraction (II)

Image removed for copyright reasons. Please see the diagram at [http://capsicum.me.utexas.edu/ChE386K/html/powder\\_diffraction\\_3.htm](http://capsicum.me.utexas.edu/ChE386K/html/powder_diffraction_3.htm).

# Powder diffraction (III)

Diagrams of the Powder Method removed for copyright reasons.  
See the images at [http://www.matter.org.uk/diffraction/x-ray/powder\\_method.htm](http://www.matter.org.uk/diffraction/x-ray/powder_method.htm)

# X-ray filters

Image removed for copyright reasons.

Please see the graph at [http://capsicum.me.utexas.edu/ChE386K/html/absorption\\_edge.htm](http://capsicum.me.utexas.edu/ChE386K/html/absorption_edge.htm).

Image removed for copyright reasons.

Please see the diagrams at <http://capsicum.me.utexas.edu/ChE386K/html/filter.htm>.

# Debye-Scherrer camera

Photographs of a Debye-Scherrer camera removed for copyright reasons.

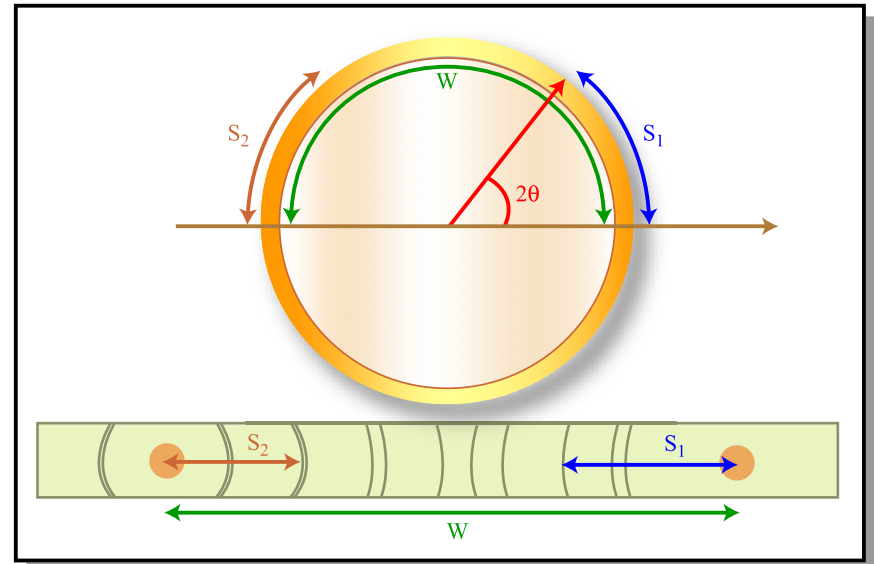


Figure by MIT OCW.



# Interplanar spacings

*Cubic:* 
$$\frac{1}{d^2} = \frac{h^2 + k^2 + l^2}{a^2}$$

*Tetragonal:* 
$$\frac{1}{d^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2}$$

*Hexagonal:* 
$$\frac{1}{d^2} = \frac{4}{3} \left( \frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}$$

*Rhombohedral:*

$$\frac{1}{d^2} = \frac{(h^2 + k^2 + l^2) \sin^2 \alpha + 2(hk + kl + hl)(\cos^2 \alpha - \cos \alpha)}{a^2(1 - 3 \cos^2 \alpha + 2 \cos^3 \alpha)}$$

*Orthorhombic:* 
$$\frac{1}{d^2} = \frac{h^2}{a^2} + \frac{k^2}{b^2} + \frac{l^2}{c^2}$$

*Monoclinic:* 
$$\frac{1}{d^2} = \frac{1}{\sin^2 \beta} \left( \frac{h^2}{a^2} + \frac{k^2 \sin^2 \beta}{b^2} + \frac{l^2}{c^2} - \frac{2hl \cos \beta}{ac} \right)$$

*Triclinic:* 
$$\frac{1}{d^2} = \frac{1}{V^2} (S_{11}h^2 + S_{22}k^2 + S_{33}l^2 + 2S_{12}hk + 2S_{23}kl + 2S_{13}hl)$$

In the equation for triclinic crystals

$$V = abc \sqrt{1 - \cos^2 \alpha - \cos^2 \beta - \cos^2 \gamma + 2 \cos \alpha \cos \beta \cos \gamma}$$

$$S_{11} = b^2 c^2 \sin^2 \alpha,$$

$$S_{22} = a^2 c^2 \sin^2 \beta,$$

$$S_{33} = a^2 b^2 \sin^2 \gamma,$$

$$S_{12} = abc^2 (\cos \alpha \cos \beta - \cos \gamma),$$

$$S_{23} = a^2 bc (\cos \beta \cos \gamma - \cos \alpha),$$

$$S_{13} = ab^2 c (\cos \gamma \cos \alpha - \cos \beta).$$

**Cubic:** 
$$d_{hkl}^2 = \frac{a^2}{h^2 + k^2 + l^2}$$

# Debye-Scherrer camera

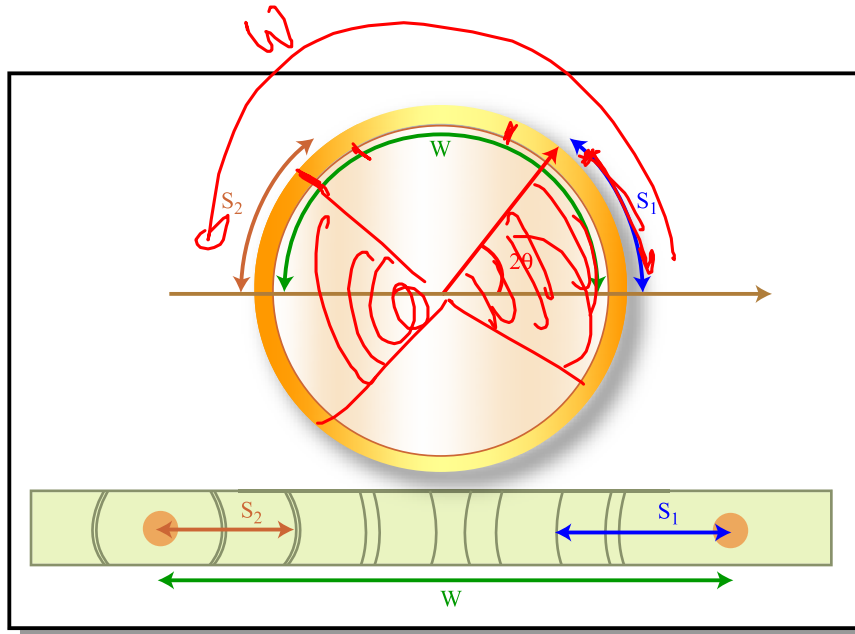


Figure by MIT OCW.

$$\lambda = d_{hkl} 2 \sin \theta$$

$$\text{Cubic: } d_{hkl}^2 = \frac{a^2}{h^2 + k^2 + l^2}$$

$$\sin \theta = \frac{\lambda}{2 d_{hkl}}$$

$$\sin^2 \theta = \frac{\lambda^2}{4} \frac{h^2 + k^2 + l^2}{a^2}$$

$$2\theta = \frac{s_1}{w} \text{ RADIANS}$$

Tables removed for copyright reasons. See [http://www.matter.org.uk/diffraction/x-ray/indexing\\_powder\\_pattern.htm](http://www.matter.org.uk/diffraction/x-ray/indexing_powder_pattern.htm)

PLANES  
PRESENT  
IN CRYSTAL

# Systematic absences

Image removed for copyright reasons.

Please see the table at [http://capsicum.me.utexas.edu/ChE386K/html/systematic\\_absences.htm](http://capsicum.me.utexas.edu/ChE386K/html/systematic_absences.htm).

# Effects of symmetry on diffraction

Images removed for copyright reasons.

Please see the images at [http://capsicum.me.utexas.edu/ChE386K/html/diffraction\\_symmetry1.htm](http://capsicum.me.utexas.edu/ChE386K/html/diffraction_symmetry1.htm).

# Structure Factor

STRUCTURE FACTOR

$$F(hkl) = \sum_{n=1}^N f_n e^{2\pi i(hx_n + ky_n + lz_n)}$$

SUM ATOMS

Image removed for copyright reasons.

Please see the graph at [http://capsicum.me.utexas.edu/ChE386K/html/scattering\\_factor\\_curve.htm](http://capsicum.me.utexas.edu/ChE386K/html/scattering_factor_curve.htm)

$$\|F\|^2 = \text{INTENSITY}$$

# Friedel's law

$$F(hkl) = F^* (-h, -k, -l)$$
$$\|F(hkl)\|^2 = \|F^* (-h, -k, -l)\|^2$$

- The diffraction pattern is always centrosymmetric, even if the crystal is not centrosymmetric

# Point symmetry + inversion = Laue

Image removed for copyright reasons.

Please see the table at [http://capsicum.me.utexas.edu/ChE386K/html/diffraction\\_symmetry2.htm](http://capsicum.me.utexas.edu/ChE386K/html/diffraction_symmetry2.htm).



# Back-reflection and transmission Laue

Diagrams of the Laue Method removed for copyright reasons.  
See the images at [http://www.matter.org.uk/diffraction/x-ray/laue\\_method.htm](http://www.matter.org.uk/diffraction/x-ray/laue_method.htm).